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MOSFET on Chemical-Mechanical Planarized Heteroepitaxial Diamond



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875-MW/cm² Low-Resistance NO₂ p-Type Doped Chemical Mechanical Planarized Diamond MOSFETs

Niloy Chandra Saha[®], Seong-Woo Kim[®], Toshiyuki Oishi[®], *Senior Member, IEEE*, and Makoto Kasu

Abstract—In this study, an Al₂O₃ passivated, NO₂ p-type doped diamond metal–oxide–semiconductor field-effect transistor (MOSFET) was fabricated on a chemical mechanical planarized high-quality heteroepitaxial diamond (Kenzan diamond[®]) substrate. This MOSFET exhibited a low specific ON-resistance of 7.54 m $\Omega \cdot$ cm² and a high OFF-state breakdown voltage of –2568 V. The chemical mechanical planarization (CMP) was performed for 200 h on the diamond surface which effectively removed the subsurface damages resulting in a low resistive diamond surface. Thus, the MOSFET showed a high drain current density of –0.68 A/mm and a maximum available power density (Baliga's figure-of-merit) of 874.6 MW/cm²—the highest reported value for diamond devices.

Index Terms— CMP, diamond MOSFET, heteroepitaxial diamond, high BFOM, NO₂ p-type doping.

I. INTRODUCTION

D IAMOND, with an ultrawide bandgap of 5.47 eV, is the most promising semiconductor material for high-power and high-frequency transistors. It possesses a high breakdown electric field of >10 MV/cm [1], [2] and a thermal conductivity of 22 W/cm·K [3]. Further, undoped diamond exhibits high carrier mobilities (electron and hole mobilities of 4500 and $3800 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$, respectively) [4]. Hydrogen termination on diamond forms a two-dimensional hole channel, and the hole density and mobility can be controlled by varying the temperature during the hydrogen termination [5]. Hydrogen terminated diamond (H-diamond) field-effect transistors show excellent radiofrequency (RF) characteristics, such as RF power densities of 2.1 and 3.8 W/mm at 1 GHz [6], [7], a maximum cut-off frequency (f_{T}) of 70 GHz [8], and a maximum oscillation frequency (f_{max}) of 120 GHz [9].

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NO₂ p-type doping is used in H-diamond to increase the hole carrier concentration (p_s) , which can be reportedly increased by one order of magnitude (up to $\sim 2.4 \times 10^{14} \text{ cm}^{-2}$) than that in air [10], [11]. An Al₂O₃ layer is used to passivate the hole channel and attain thermal stabilization [12]. Hirama et al. [13] demonstrated a diamond metal-oxide-semiconductor field-effect transistor (MOSFET) with the highest drain current density of 1.35 A/mm, using NO₂ doping and an Al₂O₃ layer. Saha et al. [14], [15] demonstrated high-drain-current-density and high-breakdownvoltage MOSFETs on heteroepitaxial diamond, which were grown on sapphire substrates using the microneedle technique. One-inch diamond substrates with a low dislocation density $(1.4 \times 10^7 \text{ cm}^{-2})$ and high crystallinity (113.4 arcsec) were reported as well [16]. Moreover, freestanding diamond with a diameter of 2'' was demonstrated using a step-flow growth on misoriented sapphire [17]. Heteroepitaxial diamond MOSFETs also exhibited the highest lateral breakdown voltage of -2608 V and a Baliga's figure-of-merit (BFOM) of 345 MW/cm² [18].

Mechanical polishing, which induces subsurface damages [19], has been widely used on heteroepitaxially grown diamond substrates to obtain a flat surface. To eliminate the subsurface damages and obtain an atomically flat surface, which is desirable for a smoother carrier transport at the heterointerfaces, chemical mechanical planarization (CMP) is applied [20]. In this study, we fabricated NO₂ p-type doped diamond MOSFETs on heteroepitaxial diamond substrates treated using CMP for 200 h. These MOSFETs exhibited very high drain currents, high mobilities, low ON resistances, and high breakdown voltages, and therefore, a high BFOM.

II. GROWTH AND FABRICATION

Fig. 1(a) shows the cross-section of the diamond MOSFET fabricated on an $8.0 \times 8.0 \text{ mm}^2$ high-quality heteroepitaxial (001) diamond freestanding substrate (Kenzan diamond[®]). The diamond substrate was grown on sapphire using the microneedle technique and then, treated with CMP for 200 h to remove the mechanical polishing-induced subsurface damages. CMP was performed using the abrasive slurry of Cr₂O₃ at a rotating speed of 3.6 m/s and a pressure of 15 kPa. During the 50 and 200 h of CMP, approximately 90- and 360- μ m-thick diamond layers were removed, respectively. As the CMP

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Fig. 1. (a) Schematic of the cross-section and (b) microscopic top view of an NO₂ p-type doped heteroepitaxial diamond (Kenzan diamond[®]) MOSFET.

progressed, the hidden subsurface damages appeared on the surface, and 200 h of CMP was required to completely remove these damages. The full width at half maximum of the 004 X-ray rocking curve was 259 arcsec, and the surface roughness was 0.04 nm. An approximately 125-nm-thick diamond homoepitaxial layer was grown on the diamond substrate using microwave plasma chemical vapor deposition. During the growth, source gases CH_4 and H_2 were supplied in a ratio of 1:100, and the microwave power and working pressure were 750 W and 50 Torr, respectively.

The NO₂ p-type doping was performed by exposing the H-diamond sample to 2 % NO₂ gas diluted in N₂. To form the ohmic contact, a 50-nm-thick Au layer was deposited on it. The active hole channel was formed by separating the source–drain (S/D) contacts. However, during the S/D fabrication, the active channel was subjected to several processes that reduced the sheet concentration. Thus, the active channel was again exposed to 2 % NO₂ (diluted in N₂).

A 16-nm-thick Al_2O_3 bilayer, which also served as the gate insulating layer, was deposited using the atomic layer deposition method in a high vacuum chamber to passivate the channel without exposing it to air. Trimethylaluminum and H_2O were sequentially supplied as the Al and O sources, respectively. The deposition temperatures of the Al_2O_3 bilayer were 120 °C (first layer: 4-nm thick) and 230 °C (second layer: 12-nm thick), respectively.

Finally, a 50-nm-thick gate (Au layer) was deposited adjacent to the source contact, depicted as the microscopic top view of the diamond MOSFET in Fig. 1(b) [21]. The active channel length (source-to-drain length, L_{SD}), gate length (L_G), and gate width were 13.6, 2.4, and 44.9 μ m, respectively. The DC output characteristics of the MOSFETs were measured at 25 °C using the Keysight 1505A power device analyzer and Keithley 248 high voltage supply.

III. RESULTS AND DISCUSSION

Fig. 2(a) shows the current-voltage characteristics of two ohmic contacts with a spacing of 13.6 μ m at different stages of fabrications. A total resistance (R_T) of 68.5 Ω -mm was determined after the S/D fabrication and a significantly reduced R_T of 38 Ω -mm was measured when the NO₂ doping was performed again at the active channel. It indicates that increased resistance due to the fabrication-related processes can be substantially lowered by the NO₂ p-type doping. However, after the Al₂O₃ bilayer deposition, R_T becomes 53 Ω -mm.

Fig. 2(b) shows the variations in $R_{\rm T}$ with different contact spacings *d* (CMP for 200 h). The contact resistance ($R_{\rm C}$) and sheet resistance ($R_{\rm SH}$) were determined using the transfer length method (TLM) performed after the Al₂O₃



Fig. 2. (a) Current–voltage characteristics of the two ohmic contacts at different stages of fabrication, and (b) R_T as a function of the ohmic contacts spacing, d, after the Al_2O_3 bilayer deposition on the heteroepitaxial diamond (CMP of 200 h). (c) Comparison of the TLM-measured parameters corresponding to diamond substrates with CMP treatment of 0, 50 [14], and 200 h.



Fig. 3. (a) $I_D - V_{DS}$ and (b) $\mu_{eff} - V_{GS}$ characteristics of the diamond MOSFET with NO₂ p-type doping and an Al₂O₃ passivation layer.

bilayer deposition. The $R_{\rm T}$ values at different d were fitted linearly, yielding $2R_{\rm C} = 5.24 \ \Omega \cdot \rm{mm}$ as the intercept, $R_{\rm SH} = 3.55 \ \rm{k}\Omega/\rm{sq}$ as the slope, and a transfer contact length $(L_{\rm T})$ of 1.48 $\mu \rm{m}$.

Furthermore, Fig. 2 (c) presents a comparison between the TLM-measured parameters of the CMP-treated diamond substrates with treatment times of 0, 50, and 200 h. Evidently, $R_{\rm C}$, $R_{\rm SH}$, and the contact resistivity ($\rho_{\rm C}$) decrease with the increasing treatment time. The lowest $\rho_{\rm C}$ of $1.94 \times 10^{-5} \ \Omega \cdot {\rm cm}^2$ was obtained for the diamond substrates treated for 200 h. Owing to the lower resistive H-diamond surface of 200 h of CMP treated sample has lower R_C and $\rho_{\rm C}$. Even the diamond surface roughnesses after the CMP of 0, 50, and 200 h were within the same level, i.e., 0.02, 0.05, and 0.04 nm, respectively. Thus, the effective removal of subsurface damages using CMP results in a highly conductive H-diamond surface.

Fig. 3(a) shows the drain current–voltage (I_D-V_{DS}) characteristics of the diamond MOSFET on the heteroepitaxial



Fig. 4. (a) OFF-state $I_{\rm D} - V_{\rm DS}$ characteristics of the diamond MOSFETs at $V_{\rm GS} = 7$ V, and (b) ON-state and OFF-state drain current characteristics of the diamond MOSFETs at different $V_{\rm GS}$.

diamond (200 h of CMP). The maximum $I_{\rm D}$ was 0.68 A/mm at a gate voltage ($V_{\rm GS}$) of -7 V and $V_{\rm DS}$ of -40 V. A low ON resistance of 50 Ω ·mm was estimated from the linear region of the I_D curve at a $V_{\rm GS}$ of -7 V. The low $R_{\rm C}$ and $R_{\rm SH}$ enabled the MOSFET to exhibit such a high current and low resistance. The gate leakage current was $<1 \ \mu$ A/mm, and the maximum transconductance ($g_{\rm m,max} = dI_{\rm D}/dV_{\rm GS}$) was 101 mS/mm at $V_{\rm GS} = -1$ V. The threshold voltage ($V_{\rm th}$) was 4.1 ± 0.1 V, which was determined by linearly extrapolating the $I_{\rm D}-V_{\rm GS}$ characteristics at $V_{\rm DS} = -0.2$ V, indicating that the MOSFET exhibits a normal ON operation.

Fig. 3(b) shows the gate-voltage-dependent effective mobility ($\mu_{\text{eff}}-V_{\text{GS}}$) characteristics. The μ_{eff} values were estimated using (1) below:

$$R_{CH}W_G = \frac{L_G}{\mu_{eff}C_{ox}|V_{GS} - V_{th}|} \tag{1}$$

where $R_{\rm CH}$ is the channel resistance under the gate, and the capacitance of the Al₂O₃ layer, $C_{\rm OX}$, is 343 nF/cm² (determined by measuring the capacitance of the metal–insulator–metal structure). The R_{CH} values were obtained using the equation, $R_{\rm CH} = R_{\rm ON} - 2R_{\rm C} - R_{\rm SH} (L_{\rm SD} - L_{\rm G})$, and a maximum $\mu_{\rm eff}$ of 205 cm²/Vs was estimated.

Fig. 4(a) shows the OFF-state drain current characteristics of the MOSFET at $V_{\rm GS} = 7$ V, indicating a high breakdown voltage, $V_{\rm BR}$, of -2568 V. Further increase in the drain voltage produces irreversible damage to the devices. The current flowing through the gate edge on the drain side to the diamond is responsible for the OFF-state drain leakage current. The lateral breakdown field, $E_{\rm BR} (= V_{\rm BR}/L_{\rm GD})$ was estimated to be 2.3 MV/cm. The gate-to-drain length, $L_{\rm GD}$ was 11 μ m. Although this $E_{\rm BR}$ is comparable to that of GaN or SiC, it is still much lower than the theoretical value for diamond.

Fig. 4(b) shows a comparison between the ON-state and OFF-state I_D-V_{DS} characteristics at different gate biases. The ON-state breakdown voltages were -821 and -1442 V at a V_{GS} of 2 and 3 V, respectively, and the ON-state currents were started to decrease when V_{DS} reached -40 V. Hot carrier injection into the Al₂O₃ layer degrades the drain current at higher drain voltages. In contrast, the breakdown voltage of -1502 V, obtained at $V_{GS} = 5$ V, can be considered as the OFF-state, because the V_{th} was 4.1 V. A higher drain leakage current flowed from the gate edge to the diamond when V_{DS}

reached -180 V. For $V_{GS} = 7$ V, no further increase in the drain current was observed, which enabled the MOSFET to exhibit a high drain voltage of -2568 V.

The MOSFET possessed an active area consisted of L_{SD} and $2L_{T}$. The specific ON-resistance, $R_{ON,spec}$, was determined to be 7.54 m Ω ·cm², and consequently, the BFOM was obtained as 874.6 MW/cm² using the equation, BFOM = $V_{BR}^2/R_{ON,spec}$, where V_{BR} , L_{SD} , and $2L_T$ were -2568 V, 13.6 μ m, and 1.48 μ m, respectively. This BFOM value is the highest value ever reported for diamond devices. Further, this BFOM exceeds our previously reported values, because the MOSFETs prepared on the heteroepitaxial diamond substrates, which were treated using CMP for 200 h, exhibited exceptionally low resistances. Among the lateral devices, GaN high electron mobility transistors demonstrated a high-power operation, with a maximum BFOM (to the best of our knowledge) of 2093 MW/cm² [22], however, our devices feature a BFOM of approximately 40 % of this value.

IV. CONCLUSION

In this study, we fabricated diamond MOSFETs using NO₂ p-type doping and an Al₂O₃ bilayer on a heteroepitaxial diamond treated using CMP for 200 h. The TLM measurements showed a low $R_{\rm C}$ and $R_{\rm SH}$ of 2.62 Ω ·mm and 3.55 k Ω /sq, respectively. The maximum $I_{\rm D}$ of -0.68 A/mm was measured for a MOSFET with $L_{\rm G} = 2.4 \ \mu$ m and $L_{\rm SD} = 13.6 \ \mu$ m. The $R_{\rm ON}$ and $\mu_{\rm eff}$ were determined as 50 Ω .mm and 205 cm²/Vs, respectively. The MOSFET exhibited a high breakdown voltage of -2568 V, and $R_{\rm ON,spec}$ was estimated as 7.54 m Ω ·cm². Consequently, the highest BFOM of 874.6 MW/cm² was experimentally determined, indicating that CMP treatment eliminates the subsurface damages, and therefore, improves the MOSFET characteristics.

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